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**Kizilyalli et al.**

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(54) **HIGH POWER GALLIUM NITRIDE  
ELECTRONICS USING MISCUT  
SUBSTRATES**

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**H01L 21/76** (2006.01)  
**H01L 29/04** (2006.01)  
**H01L 29/06** (2006.01)

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(2013.01); **H01L 21/02389** (2013.01); **H01L**  
**21/02433** (2013.01); **H01L 21/02458** (2013.01);  
**H01L 21/7605** (2013.01); **H01L 29/045**  
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**29/66204** (2013.01); **H01L 29/861** (2013.01);  
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H01L 29/8611; C30B 25/02; C30B 25/18;  
C30B 25/00; C30B 25/16  
USPC ..... 257/E21.09, E33.003, E33.005,  
257/E21.056

See application file for complete search history.

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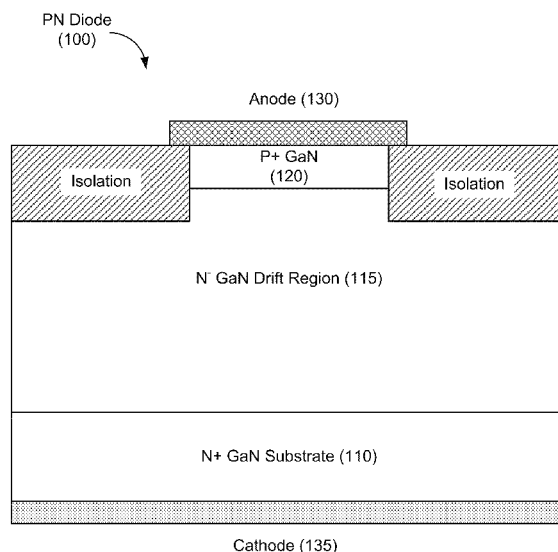
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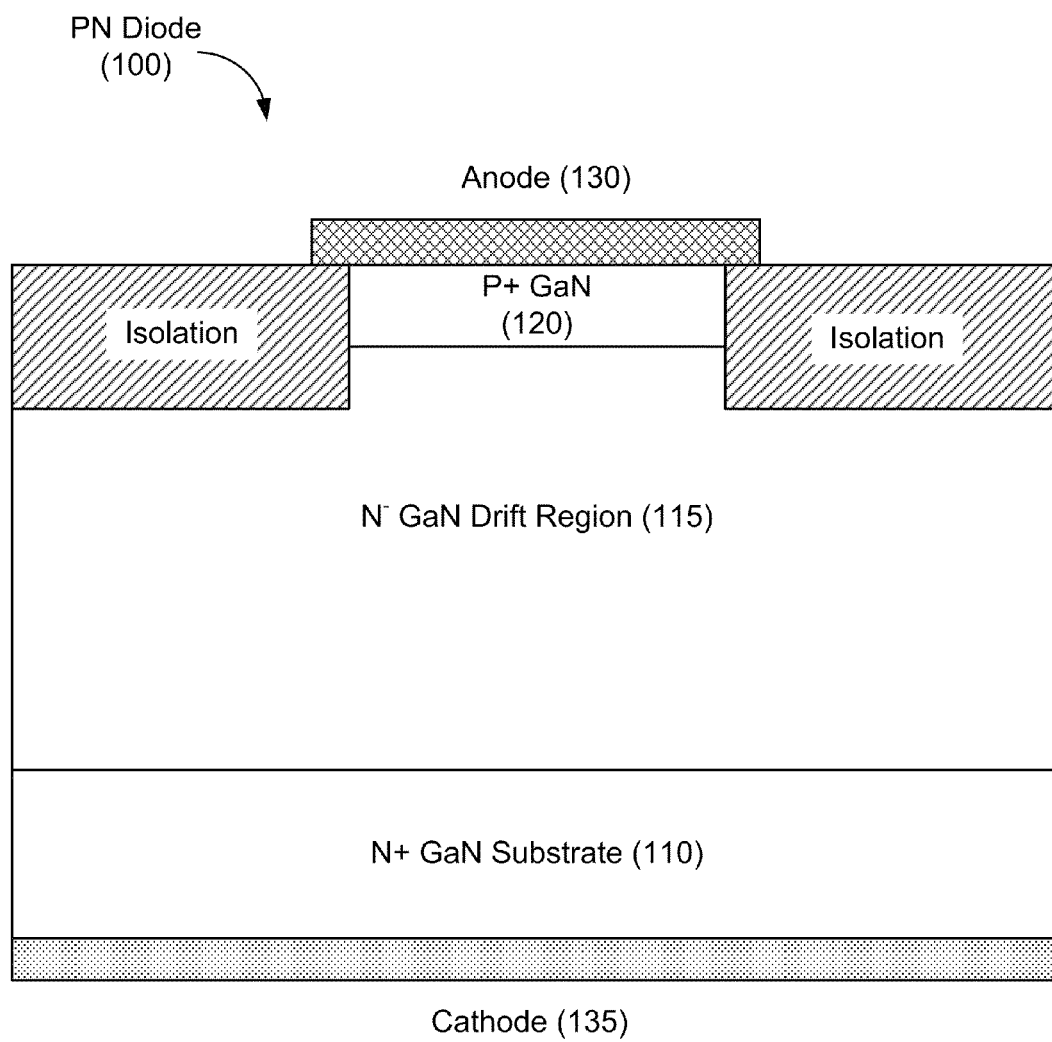
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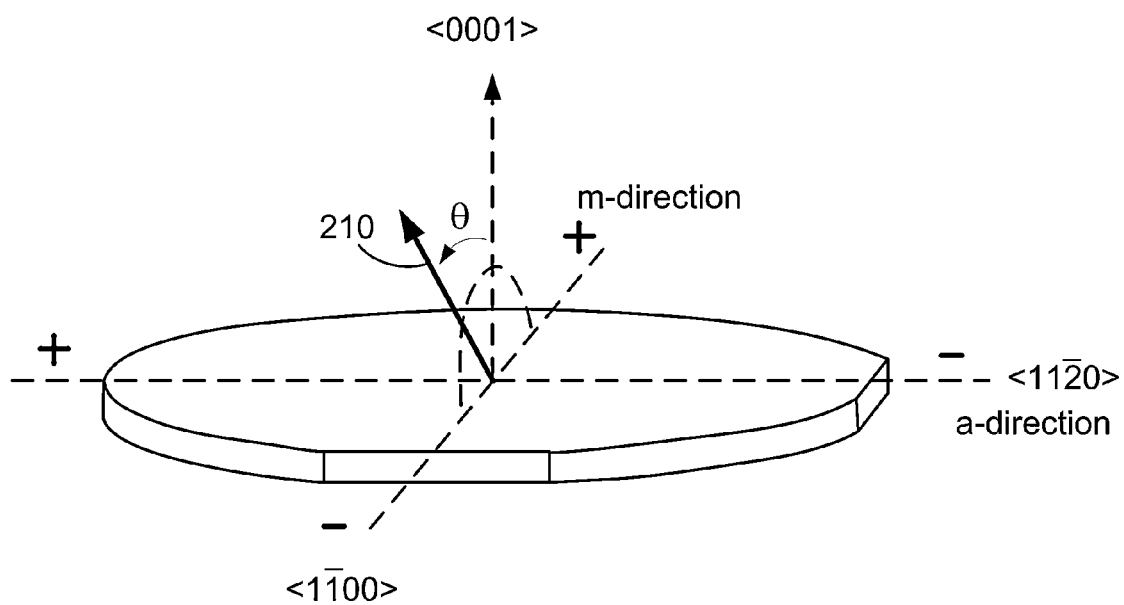
(57) **ABSTRACT**

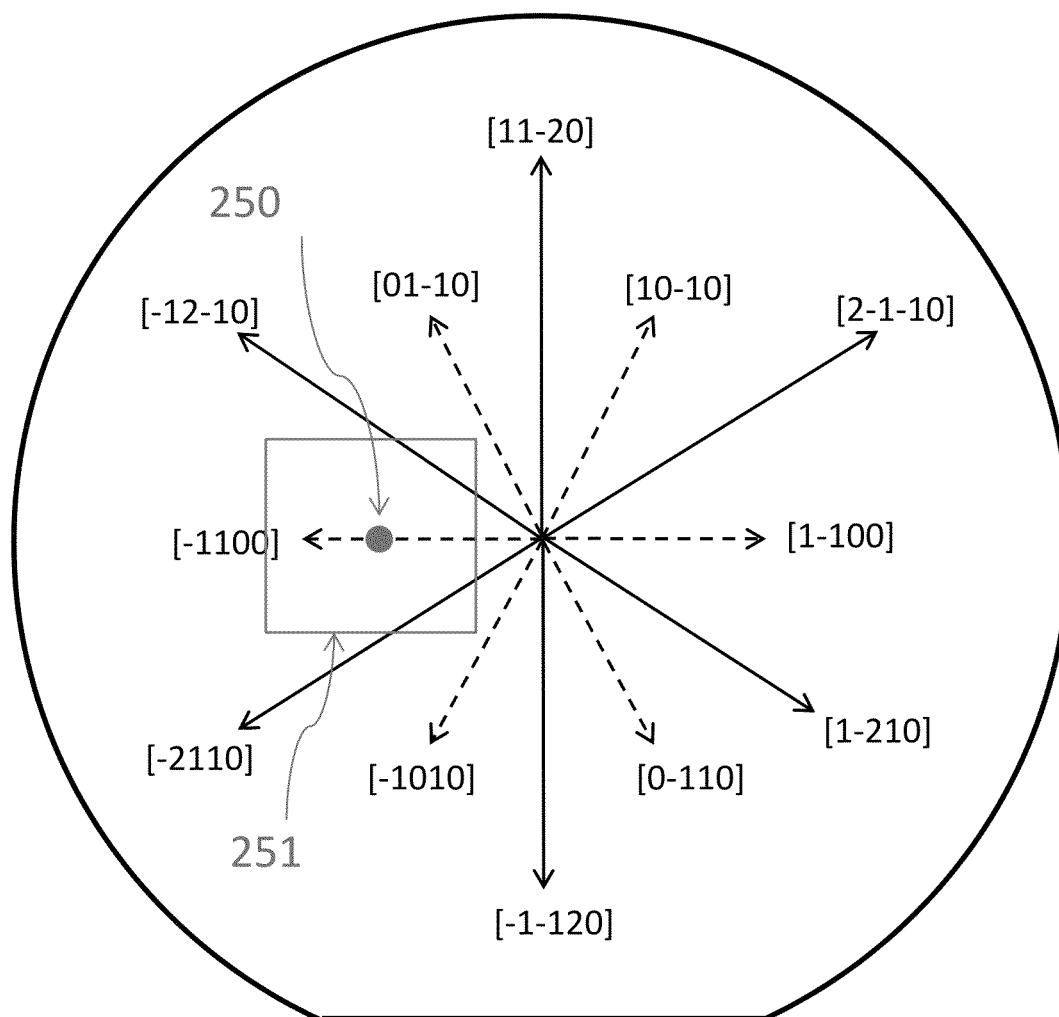
An electronic device includes a III-V substrate having a hexagonal crystal structure and a normal to a growth surface characterized by a misorientation from the <0001> direction of between 0.15° and 0.65°. The electronic device also includes a first epitaxial layer coupled to the III-V substrate and a second epitaxial layer coupled to the first epitaxial layer. The electronic device further includes a first contact in electrical contact with the substrate and a second contact in electrical contact with the second epitaxial layer.

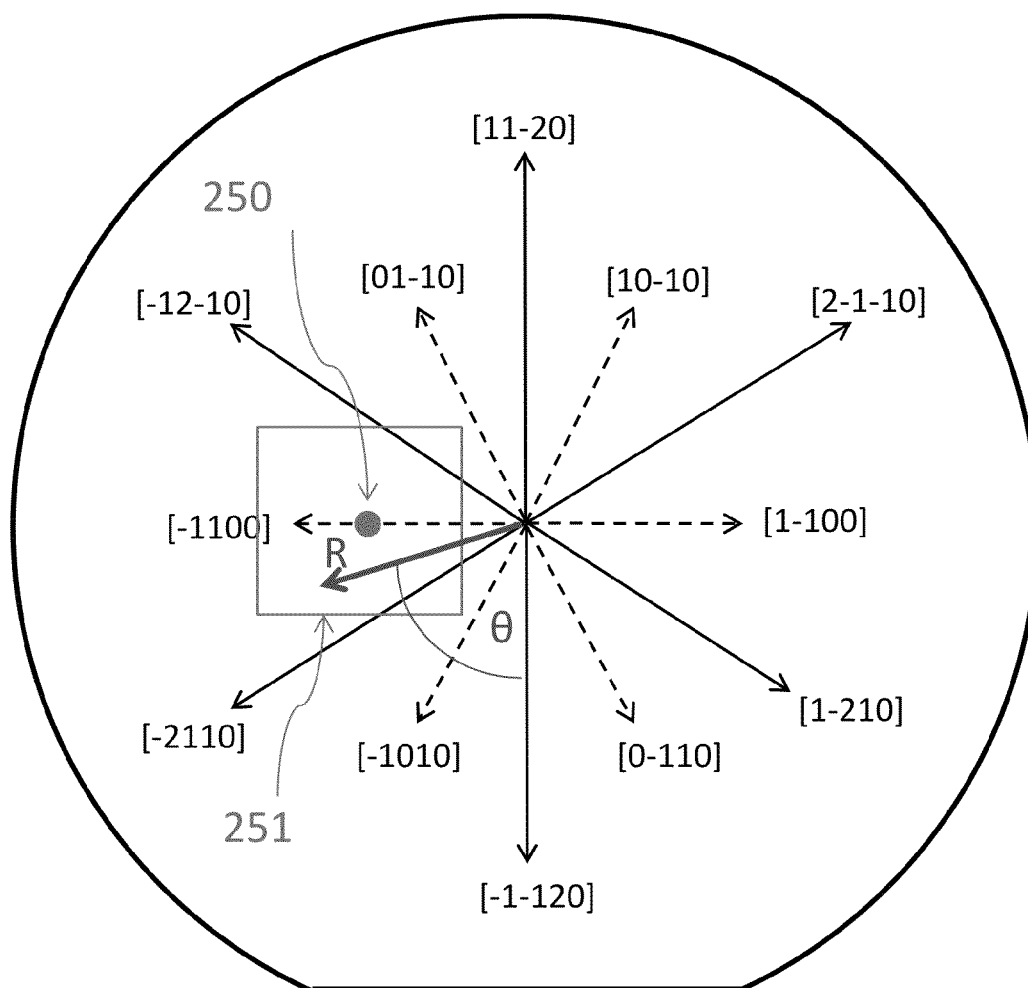
**10 Claims, 13 Drawing Sheets**

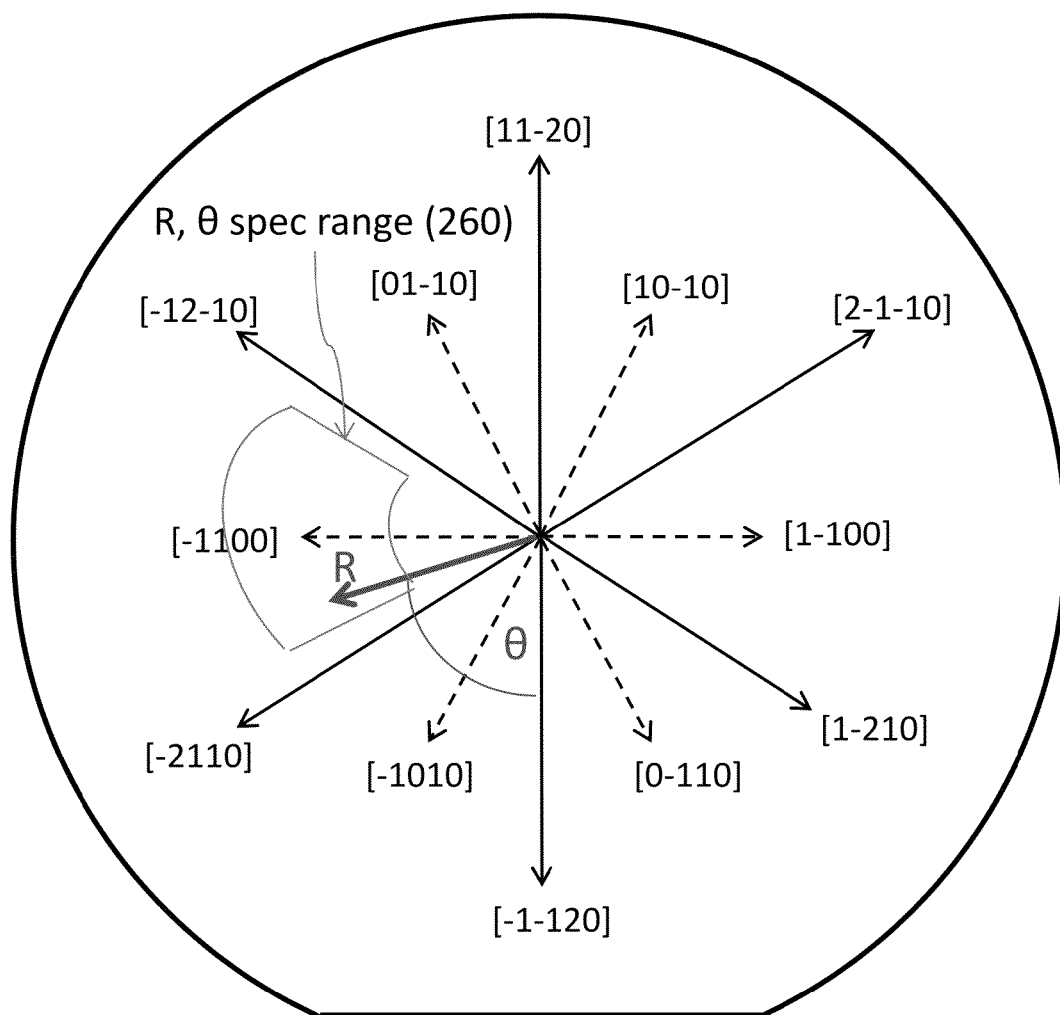


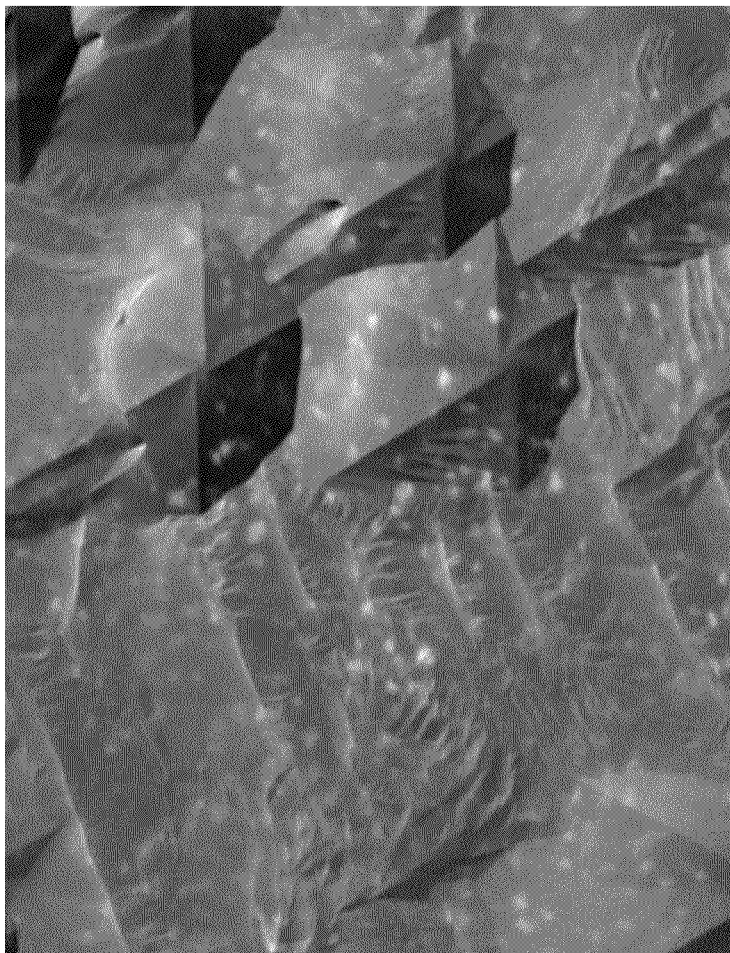
**FIG. 1**

**FIG. 2A**

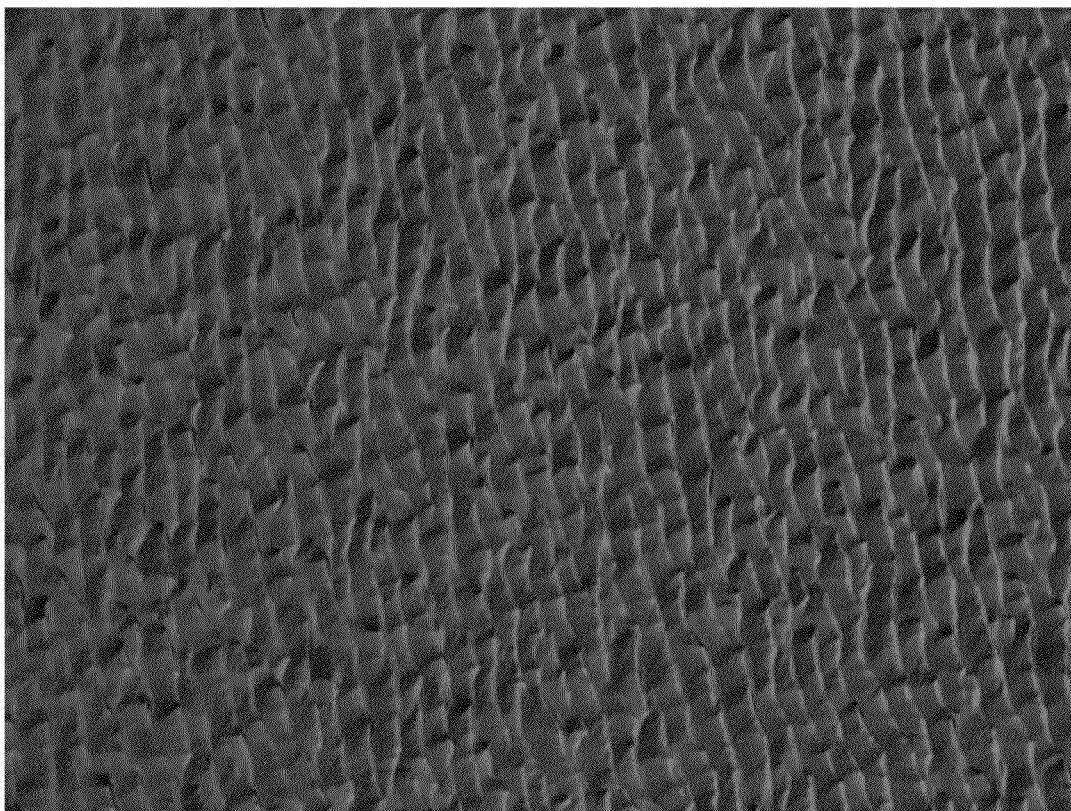
**FIG. 2B**

**FIG. 2C**

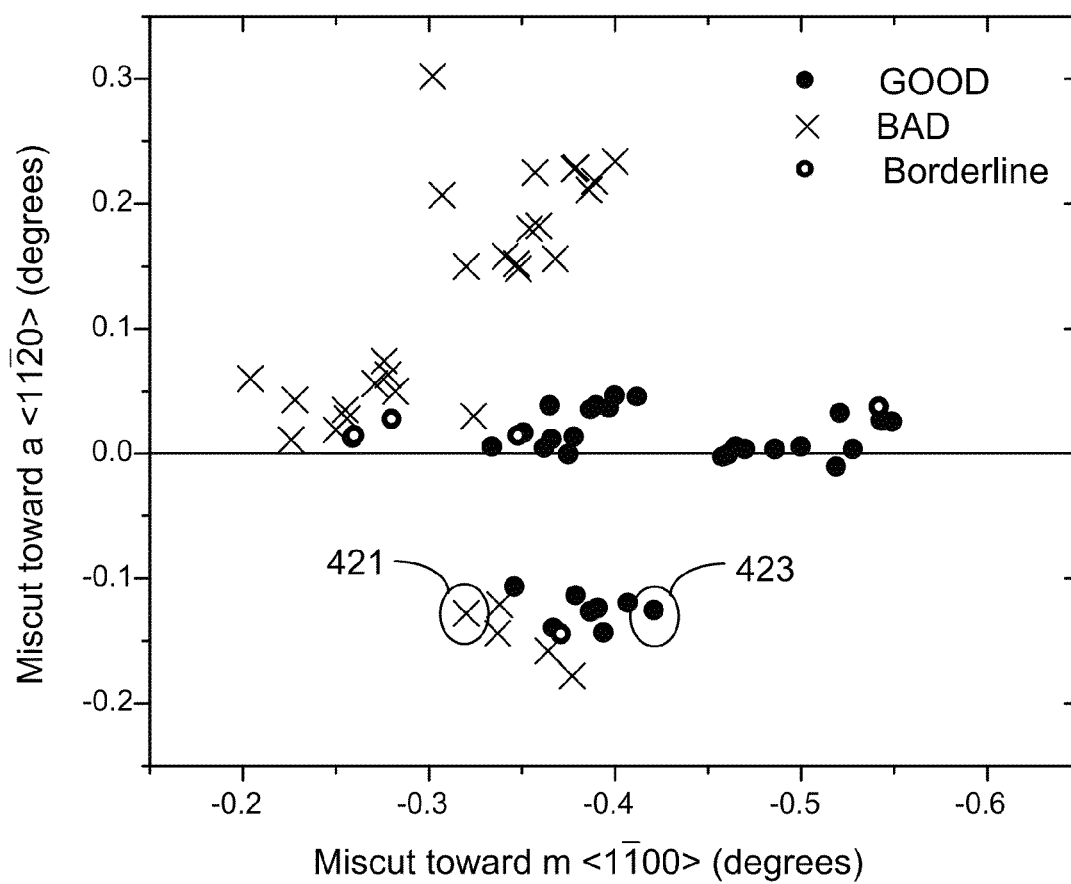
**FIG. 2D**



***FIG. 3A***



***FIG. 3B***

**FIG. 4A**

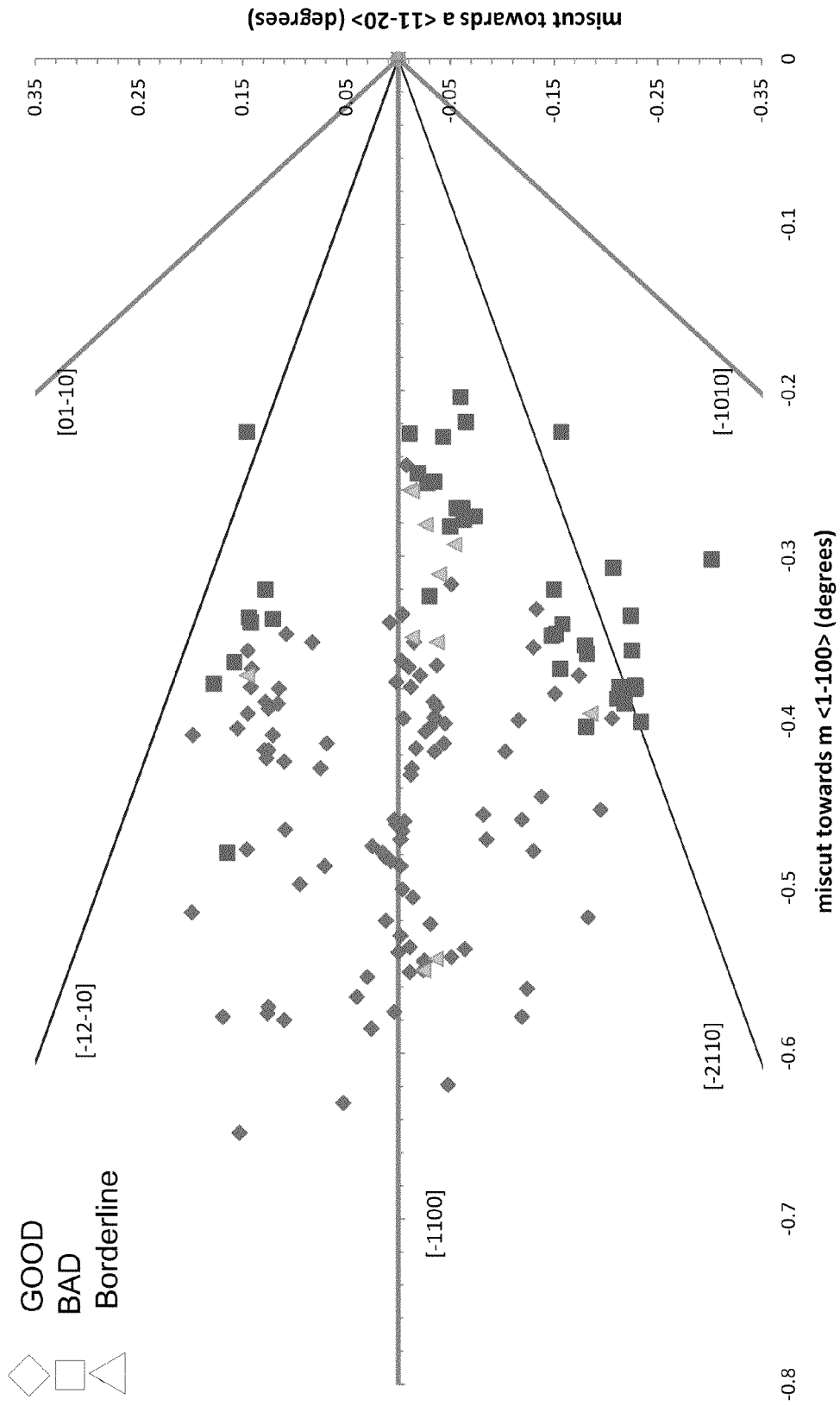
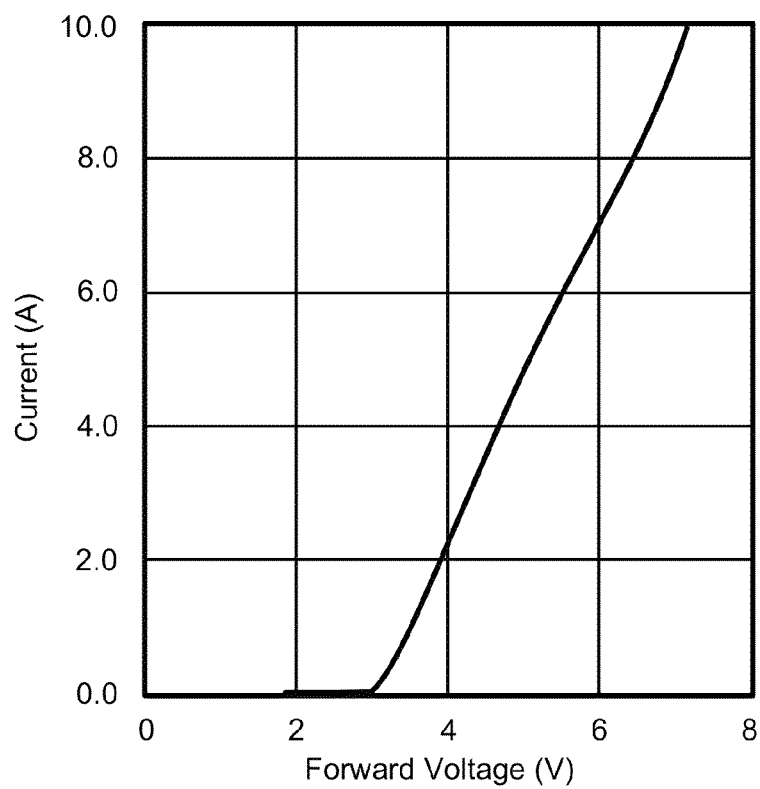
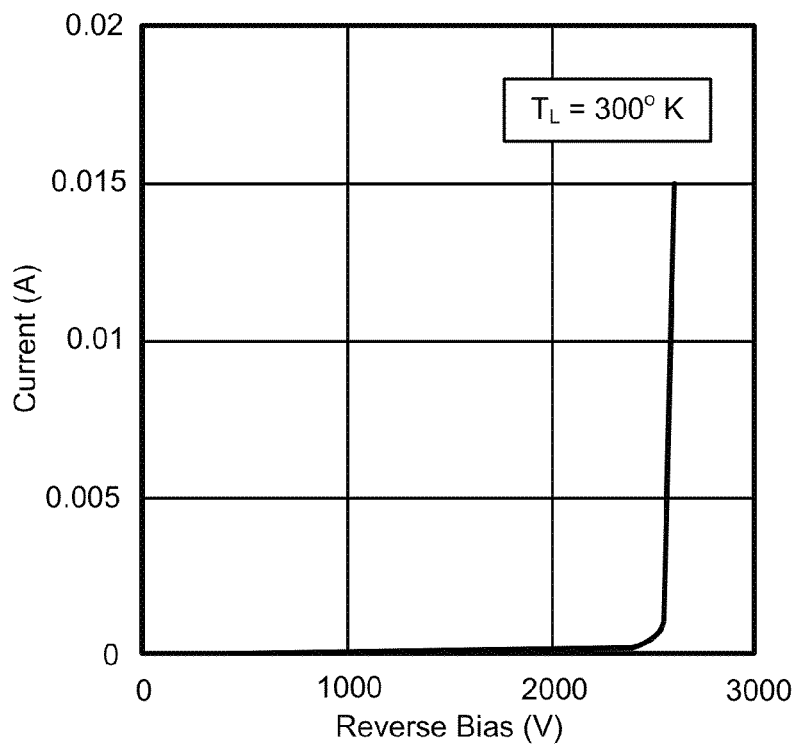
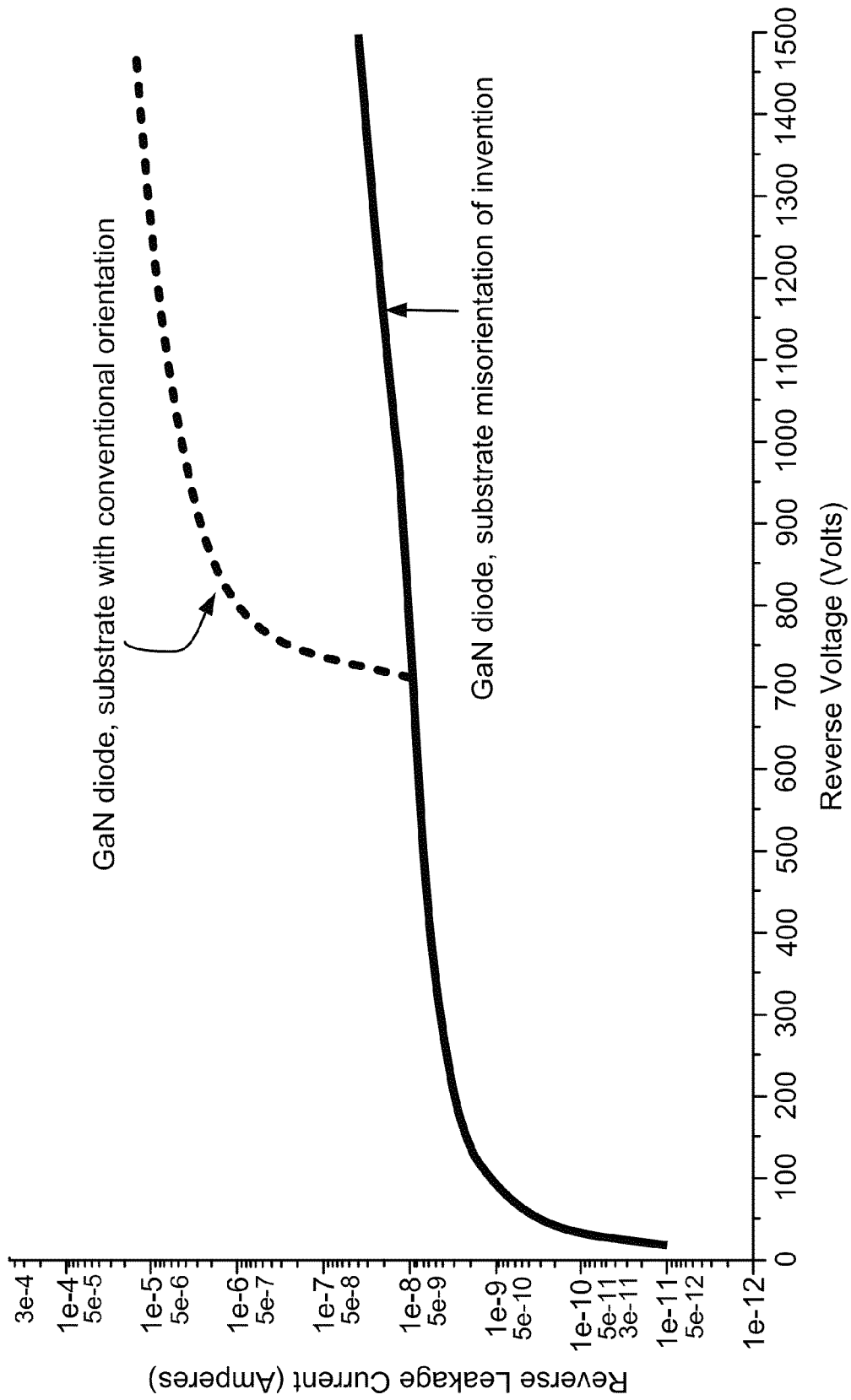
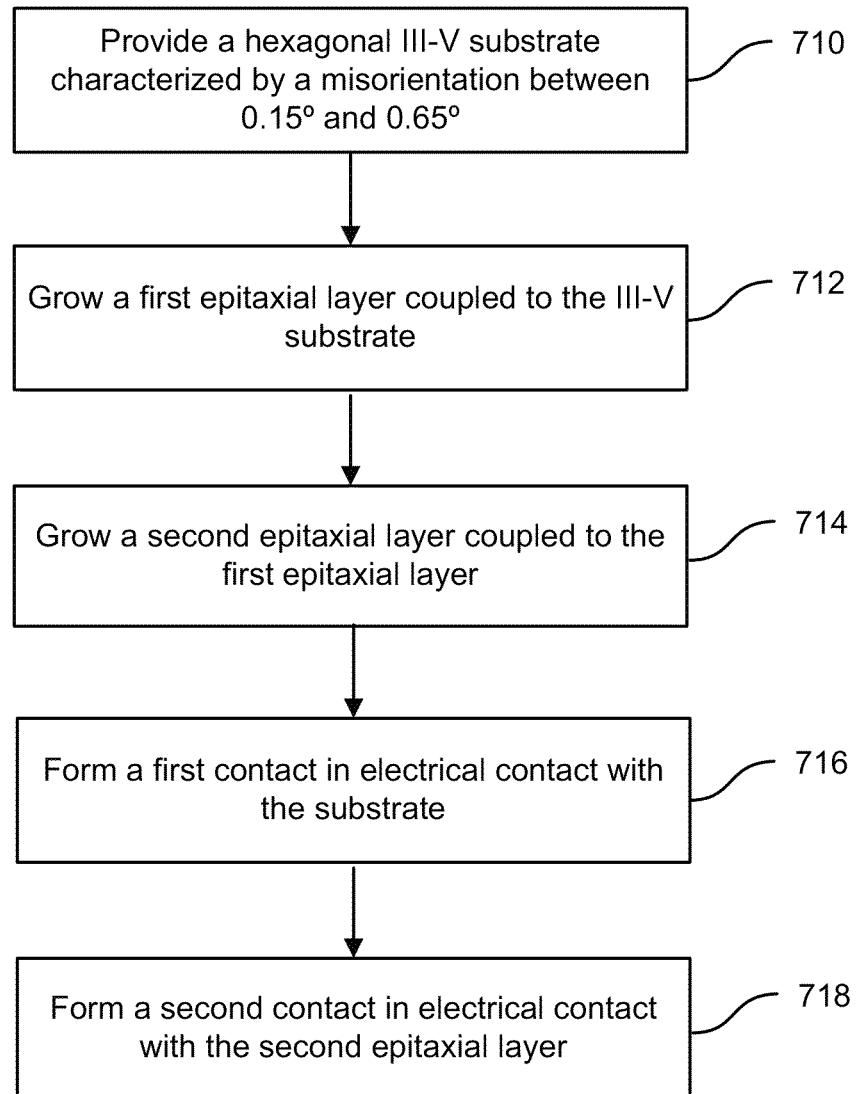
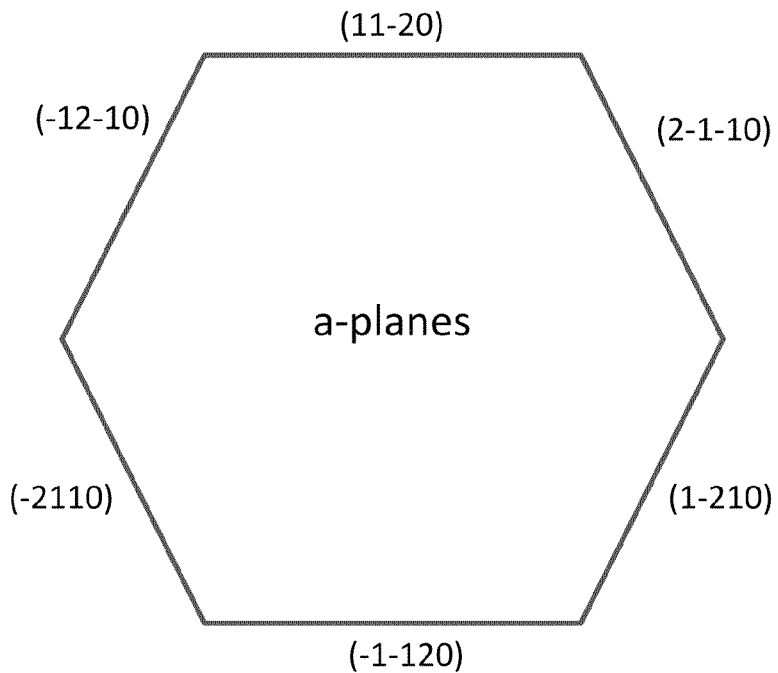
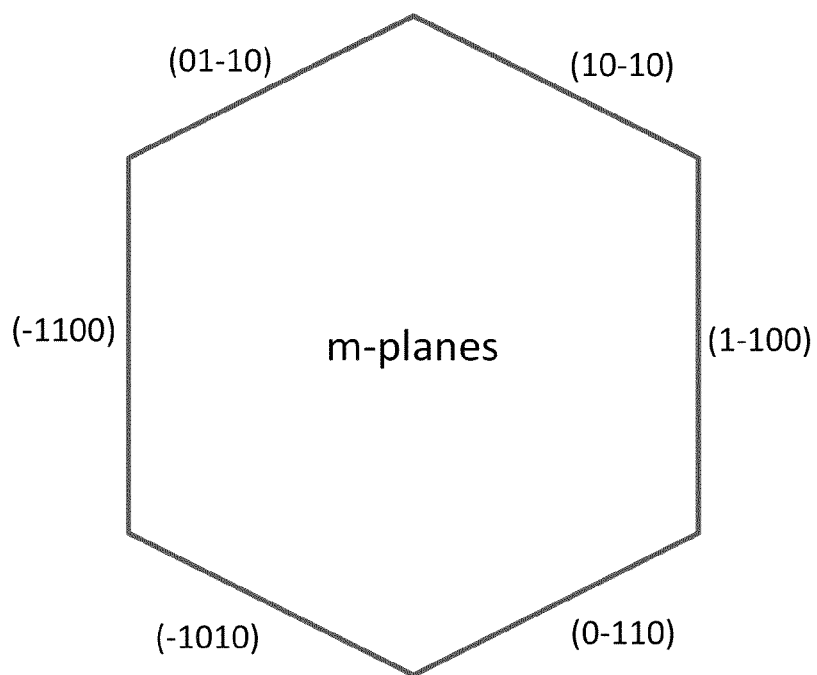


FIG. 4B

**FIG. 5A****FIG. 5B**

**FIG. 6**

**FIG. 7**

**FIG. 8A****FIG. 8B**

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# HIGH POWER GALLIUM NITRIDE ELECTRONICS USING MISCUT SUBSTRATES

## BACKGROUND OF THE INVENTION

Power electronics are widely used in a variety of applications. Power electronic devices are commonly used in circuits to modify the form of electrical energy, for example, from ac to dc, from one voltage level to another, or in some other way. Such devices can operate over a wide range of power levels, from milliwatts in mobile devices to hundreds of megawatts in a high voltage power transmission system. Despite the progress made in power electronics, there is a need in the art for improved electronics systems and methods of operating the same.

## SUMMARY OF THE INVENTION

The present invention relates generally to electronic devices. More specifically, the present invention relates to the fabrication of gallium nitride (GaN) based epitaxial layers useful for high power electronics. In a particular embodiment, a GaN substrate with the growth plane misoriented from the (0001) plane by less than one degree in relation to the  $\langle 1100 \rangle$  direction is utilized in an epitaxial growth process. The surface morphology and electrical properties of epitaxial layers grown on the misoriented substrate are suitable for use in high power electronic devices. The methods and techniques can be applied to a variety of compound semiconductor systems including diodes, FETs, and the like.

According to an embodiment of the present invention, an electronic device is provided. The electronic device includes a III-V substrate having a hexagonal crystal structure and a normal to a growth surface characterized by a misorientation from the  $\langle 0001 \rangle$  direction of between  $0.15^\circ$  and  $0.65^\circ$ . The electronic device also includes a first epitaxial layer coupled to the III-V substrate and a second epitaxial layer coupled to the first epitaxial layer. The electronic device further includes a first contact in electrical contact with the substrate and a second contact in electrical contact with the second epitaxial layer.

According to another embodiment of the present invention, a method of fabricating an electronic device is provided. The method includes providing a III-V substrate having a hexagonal crystal structure and a normal to a growth surface characterized by a misorientation from the  $\langle 0001 \rangle$  direction of between  $0.15^\circ$  and  $0.65^\circ$ . The method also includes growing a first epitaxial layer coupled to the III-V substrate and growing a second epitaxial layer coupled to the first epitaxial layer. The method further includes forming a first contact in electrical contact with the substrate and forming a second contact in electrical contact with the second epitaxial layer.

Numerous benefits are achieved by way of the present invention over conventional techniques. For example, embodiments of the present invention provide methods and systems for fabricating epitaxial layers suitable for use in high power electronics devices. In an embodiment, device performance during high voltage operation (e.g., voltages greater than 200 V) is improved in comparison with conventional designs. These and other embodiments of the invention, along with many of its advantages and features, are described in more detail in conjunction with the text below and attached figures.

## BRIEF DESCRIPTION OF THE DRAWINGS

FIG. 1 is a simplified cross-sectional diagram of a high voltage PN diode structure according to an embodiment of the present invention.

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FIG. 2A is a schematic diagram illustrating Miller indices of a hexagonal-phase bulk GaN substrate wafer.

FIG. 2B is a schematic diagram showing the crystal axis directions for a c-face GaN crystal.

FIG. 2C is a schematic diagram illustrating the vector nature of the miscut angle according to an embodiment of the present invention.

FIG. 2D is a schematic diagram illustrating the radial-vector nature of the miscut angle according to an embodiment of the present invention.

FIG. 3A is a Nomarski micrograph of epitaxial surface, for case when substrate misorientation is  $<0.15^\circ$ .

FIG. 3B is a Nomarski micrograph of epitaxial surface, for case when substrate misorientation is  $>0.65^\circ$ .

FIG. 4A is a plot showing the surface morphologies of various epitaxially grown layers mapped against substrate misorientation in orthogonal directions.

FIG. 4B is a plot showing wafer quality data as a function of miscut angle according to an embodiment of the present invention.

FIG. 5A is a plot showing the forward bias current-voltage characteristics of a high voltage PN diode according to an embodiment of the present invention.

FIG. 5B is a plot showing the reverse bias current-voltage characteristics of a high voltage PN diode according to an embodiment of the present invention.

FIG. 6 is a plot showing the reverse bias current-voltage characteristics of a high-voltage GaN PN diode according to an embodiment of the present invention compared to a GaN PN diode fabricated using a conventional substrate.

FIG. 7 is a simplified flowchart illustrating a method of fabricating an electronic device according to an embodiment of the present invention.

FIGS. 8A-8B are diagrams illustrating crystal planes of Wurtzite crystals.

## DETAILED DESCRIPTION OF SPECIFIC EMBODIMENTS

Embodiments of the present invention relate to electronic devices. More specifically, the present invention relates to the fabrication of gallium nitride (GaN) based epitaxial layers useful for high power electronics. In a particular embodiment, a GaN substrate with the growth plane misoriented from the (0001) plane by less than one degree in relation to the  $\langle 1100 \rangle$  direction is utilized in an epitaxial growth process. The surface morphology and electrical properties of epitaxial layers grown on the misoriented substrate are suitable for use in high power electronic devices. The methods and techniques can be applied to a variety of compound semiconductor systems including diodes, FETs, and the like.

GaN-based electronic and optoelectronic devices are undergoing rapid development. Desirable properties associated with GaN and related alloys and heterostructures include high bandgap energy for visible and ultraviolet light emission, favorable transport properties (e.g., high electron mobility and saturation velocity), a high breakdown field, and high thermal conductivity. According to embodiments of the present invention, gallium nitride (GaN) epitaxy on pseudo-bulk GaN substrates is utilized to fabricate vertical GaN-based semiconductor devices not possible using conventional techniques. For example, conventional methods of growing GaN include using a foreign substrate such as silicon carbide (SiC). This can limit the thickness of a usable GaN layer grown on the foreign substrate due to differences in thermal expansion coefficients and lattice constant between the GaN layer and the foreign substrate. High defect densities at the

interface between GaN and the foreign substrate further complicate attempts to create vertical devices, including power electronic devices such as JFETs and other field-effect transistors.

Homoepitaxial GaN layers on bulk GaN substrates, on the other hand, are utilized in the embodiments described herein to provide superior properties to conventional techniques and devices. For instance, electron mobility,  $\mu$ , is higher for a given background doping level,  $N$ . This provides low resistivity,  $\rho$ , because resistivity is inversely proportional to electron mobility, as provided by equation (1):

$$\rho = \frac{1}{q\mu N}, \quad (1)$$

where  $q$  is the elementary charge.

Another superior property provided by homoepitaxial GaN layers on bulk GaN substrates is high critical electric field for avalanche breakdown. A high critical electric field allows a larger voltage to be supported over smaller length,  $L$ , than a material with a lower critical electric field. A smaller length for current to flow together with low resistivity give rise to a lower resistance,  $R$ , than other materials, since resistance can be determined by the equation:

$$R = \frac{\rho L}{A}, \quad (2)$$

where  $A$  is the cross-sectional area of the channel or current path.

In general, a tradeoff exists between the physical dimension of a device needed to support high voltage in a device's off-state and the ability to pass current through the same device with low resistance in the on-state. In many cases GaN is preferable over other materials in minimizing this tradeoff and maximizing performance. In addition, GaN layers grown on bulk GaN substrates have low defect density compared to layers grown on mismatched substrates. The low defect density will give rise to superior thermal conductivity, less trap-related effects such as dynamic on-resistance, and better reliability.

FIG. 1 is a simplified cross-sectional diagram of a high voltage PN diode structure according to an embodiment of the present invention. Referring to FIG. 1, a first gallium-nitride (GaN) epitaxial layer **115** (e.g., a  $N^-$  GaN drift region) is formed on a GaN substrate **110** having the same conductivity type. The GaN substrate **110** can be a pseudo-bulk or bulk GaN material on which the first GaN epitaxial layer **115** is grown. A buffer layer (not shown) can be utilized as will be evident to one of skill in the art. Dopant concentrations (e.g., doping density) of the GaN substrate **110** can vary, depending on desired functionality. For example, a GaN substrate **110** can have an  $n+$  conductivity type, with dopant concentrations ranging from  $1 \times 10^{17} \text{ cm}^{-3}$  to  $1 \times 10^{19} \text{ cm}^{-3}$ . Although the GaN substrate **110** is illustrated as including a single material composition, multiple layers can be provided as part of the substrate. Moreover, adhesion, buffer, and other layers (not illustrated) can be utilized during the epitaxial growth process. One of ordinary skill in the art would recognize many variations, modifications, and alternatives.

The properties of the first GaN epitaxial layer **115** can also vary, depending on desired functionality. The first GaN epitaxial layer **115** can serve as a drift region for the PN diode, and therefore can be a relatively low-doped material. For

example, the first GaN epitaxial layer **115** can have an  $n$ -conductivity type, with dopant concentrations ranging from  $1 \times 10^{14} \text{ cm}^{-3}$  to  $1 \times 10^{18} \text{ cm}^{-3}$ . Furthermore, the dopant concentration can be uniform, or can vary, for example, as a function of the thickness of the drift region.

The thickness of the first GaN epitaxial layer **115** can also vary substantially, depending on the desired functionality. As discussed above, homoepitaxial growth can enable the first GaN epitaxial layer **115** to be grown far thicker than layers formed using conventional methods. In general, in some embodiments, thicknesses can vary between  $0.5 \mu\text{m}$  and  $100 \mu\text{m}$ , for example. In other embodiments thicknesses are greater than  $5 \mu\text{m}$ . Resulting parallel plane breakdown voltages for the PN diode **100** can vary depending on the embodiment. Some embodiments provide for breakdown voltages of at least  $100\text{V}$ ,  $300\text{V}$ ,  $600\text{V}$ ,  $1.2 \text{ kV}$ ,  $1.7 \text{ kV}$ ,  $3.3 \text{ kV}$ ,  $5.5 \text{ kV}$ ,  $13 \text{ kV}$ , or  $20 \text{ kV}$ .

Referring once again to FIG. 1, a second GaN epitaxial layer **120** is formed above the first GaN epitaxial layer **115**. The second GaN epitaxial layer **120** is used in forming the  $p$ -type region of the PN diode and has a conductivity type different than the first GaN epitaxial layer **115**. For instance, if the first GaN epitaxial layer **115** is formed from an  $n$ -type GaN material, the second GaN epitaxial layer **120** will be formed from a  $p$ -type GaN material, and vice versa. As illustrated in FIG. 1, isolation regions are formed to define the lateral extent of the PN diode. Suitable techniques for forming the isolation regions characterized by high resistivity can include ion implantation, etching and epitaxial regrowth of low conductivity material, etching and deposition of insulating materials such as oxides and/or nitrides, combinations thereof, or the like. One of ordinary skill in the art would recognize many variations, modifications, and alternatives.

The thickness of the second GaN epitaxial layer **120** can vary, depending on the process used to form the layer and the device design. In some embodiments, the thickness of the second GaN epitaxial layer **120** is between  $0.1 \mu\text{m}$  and  $5 \mu\text{m}$ . In other embodiments, the thickness of the second GaN epitaxial layer **120** is between  $0.3 \mu\text{m}$  and  $1 \mu\text{m}$ .

The second GaN epitaxial layer **120** can be highly doped, for example in a range from about  $5 \times 10^{17} \text{ cm}^{-3}$  to about  $1 \times 10^{19} \text{ cm}^{-3}$ . Additionally, as with other epitaxial layers, the dopant concentration of the second GaN epitaxial layer **120** can be uniform or non-uniform as a function of thickness. In some embodiments, the dopant concentration increases with thickness, such that the dopant concentration is relatively low near the first GaN epitaxial layer **115** and increases as the distance from the first GaN epitaxial layer **115** increases. Such embodiments provide higher dopant concentrations at the top of the second GaN epitaxial layer **120** where metal contacts can be subsequently formed. Other embodiments utilize heavily doped contact layers (not shown) to form ohmic contacts.

One method of forming the second GaN epitaxial layer **120**, and other layers described herein, can be through a regrowth process that uses an in-situ etch and diffusion preparation processes. These preparation processes are described more fully in U.S. patent application Ser. No. 13/198,666, filed on Aug. 4, 2011, the disclosure of which is hereby incorporated by reference in its entirety.

FIG. 1 also illustrates electrical contacts formed for the electronic device according to an embodiment of the present invention. As illustrated in FIG. 1, a metallic structure **135** is formed in electrical contact with the GaN substrate **110**. The metallic structure **135** can be one or more layers of ohmic metal that serve as a contact for the cathode of the PN diode **100**. For example, the metallic structure **135** can comprise a

titanium-aluminum (Ti/Al) ohmic metal. Other metals and/or alloys can be used including, but not limited to, aluminum, nickel, gold, combinations thereof, or the like. In some embodiments, an outermost metal of the metallic structure 135 can include gold, tantalum, tungsten, palladium, silver, or aluminum, combinations thereof, and the like. The metallic structure 135 can be formed using any of a variety of methods such as sputtering, evaporation, or the like.

FIG. 1 also illustrates an additional metallic structure 130 in electrical contact with the second epitaxial layer 120. The additional metallic structure 130 can be one or more layers of ohmic metal including metals and/or alloys similar to the metallic structure 135. The additional metallic structure 130 is formed on the second epitaxial layer 120 to serve as the anode contact of the PN diode 100. The additional metallic structure 130 can be formed using a variety of techniques, including lift-off and/or deposition with subsequent etching, which can vary depending on the metals used. Example metals include nickel-gold (Ni/Au), and the like. In some implementations, the additional metallic structure 130 is formed in contact with the first epitaxial layer and a Schottky metal is utilized as appropriate to the formation of a Schottky diode.

Different dopants can be used to create n- and p-type GaN epitaxial layers and structures disclosed herein. For example, n-type dopants can include silicon, oxygen, or the like. P-type dopants can include magnesium, beryllium, calcium zinc, or the like.

Although some embodiments are discussed in terms of GaN substrates and GaN epitaxial layers, the present invention is not limited to these particular binary III-V materials and is applicable to a broader class of III-V materials, in particular III-nitride materials. Additionally, although a GaN substrate is illustrated in FIG. 1, embodiments of the present invention are not limited to GaN substrates. Other III-V materials, in particular, III-nitride materials, are included within the scope of the present invention and can be substituted not only for the illustrated GaN substrate, but also for other GaN-based layers and structures described herein. As examples, binary III-V (e.g., III-nitride) materials, ternary III-V (e.g., III-nitride) materials such as InGaN and AlGaIn, quaternary III-nitride materials, such as AlInGaIn, doped versions of these materials, and the like are included within the scope of the present invention.

The fabrication process discussed in relation to FIG. 1 utilize a process flow in which an n-type drift layer is grown using an n-type substrate. However, the present invention is not limited to this particular configuration. In other embodiments, substrates with p-type doping are utilized. Additionally, embodiments can use materials having an opposite conductivity type to provide devices with different functionality. Thus, although some examples relate to the growth of n-type GaN epitaxial layer(s) doped with silicon, in other embodiments the techniques described herein are applicable to the growth of highly or lightly doped material, p-type material, material doped with dopants in addition to or other than silicon such as Mg, Ca, Be, Ge, Se, S, O, Te, and the like. The substrates discussed herein can include a single material system or multiple material systems including composite structures of multiple layers. One of ordinary skill in the art would recognize many variations, modifications, and alternatives.

During the growth of the epitaxial layers illustrated in FIG. 1, the inventors have determined that epitaxial layers grown on substrates characterized by predetermined miscut angles provide improved performance in the context of high power electronic devices (e.g., operation at high voltages) in comparison with conventional structures. For epitaxial layers grown on the c-plane of a GaN substrate, the layer morphol-

ogy, and importantly, the performance of devices formed using these epitaxial layers, degrades at higher voltages, reducing their applicability to high power applications. The inventors have determined that, without limiting embodiments of the present invention, misorientation of the substrate growth surface from the crystallographic planes (e.g., the c-plane) by a fraction of a degree in a predetermined direction, improves layer morphology and device performance for devices fabricated using such improved layers.

FIGS. 8A-8B are diagrams illustrating crystal planes of Wurtzite crystals. In FIG. 8A, the a-planes of a Wurtzite crystal are illustrated and in FIG. 8B, the m-planes of a Wurtzite crystal are illustrated. The c-axis (0001) is normal to the plane of the figures and the (000-1) axis points into the plane of the figures. As illustrated in FIG. 8A, there are six a-planes, all 60° apart. As illustrated in FIG. 8B, there are six m-planes, all 60° apart. When overlaid, the m- and a-planes interpenetrate, an angle of 30° between these planes. As will be evident to one of skill in the art, GaN has a Wurtzite crystalline structure.

FIG. 2A is a schematic diagram illustrating Miller indices of a hexagonal-phase bulk GaN substrate wafer. The dashed arrows indicate the direction of the <0001>, <1 $\bar{1}$ 00>, and <1 $\bar{1}$ 20> directions. The solid arrow 210 indicates the direction of misorientation with respect to the <1 $\bar{1}$ 00> direction utilized for the epitaxial growth of the high-voltage electronic device structures described according to some embodiments herein. As illustrated in FIG. 2A, embodiments of the present invention utilize substrates in which the growth plane is not aligned with the (0001) plane. As described herein, the normal to the growth plane is misoriented from the <0001> direction (i.e., a misorientation angle ( $\theta$ )) by  $0 < \theta < 1.0^\circ$  towards the <1 $\bar{1}$ 00> direction or the <1 $\bar{1}$ 20> direction. According to some embodiments of the present invention, the magnitude of  $\theta$  ranges from about  $0.15^\circ < \theta < 0.65^\circ$ . In a particular embodiment, the misorientation angle  $\theta$  ranges from about  $0.4^\circ < \theta < 0.5^\circ$ .

Thus, according to embodiments of the present invention, the growth plane of the III-V (e.g., GaN) substrate is misoriented from the c-plane towards the positive or negative m-direction at an angle having a value between zero and  $1.0^\circ$ . Additionally, the normal to the substrate growth surface can also be misoriented such that it also tilts towards or away from the a-direction. The misorientation away from the <0001> direction towards the a-direction is zero in some embodiments. In the embodiment illustrated in FIG. 2A, the normal to the growth surface is characterized by a misorientation from the <0001> direction toward the <1 $\bar{1}$ 00> direction of between  $-0.15^\circ$  and  $-0.65^\circ$  and a misorientation from the <0001> direction toward the <1 $\bar{1}$ 20> direction of zero.

In some embodiments, the orientation of the growth surface is such that the growth surface is tilted with respect to the (0001) plane, resulting in the normal to the growth surface being tilted by less than one degree toward the positive <1 $\bar{1}$ 00> direction. The inventors have determined that tilting of the growth surface by misorientation away from the (0001) surface towards the negative m-direction by between  $0.15^\circ$  and  $0.65^\circ$  or towards the positive m-direction by between  $0.15^\circ$  and  $0.65^\circ$  results in improvements in the surface morphology for thick epitaxial layers and device performance accordingly. Thus, embodiments of the present invention provide growth surfaces tilted with respect to the (0001) surface by an angle greater than  $0.15^\circ$  and less than one degree. One of ordinary skill in the art would recognize many variations, modifications, and alternatives.

FIG. 2B is a schematic diagram illustrating miscut angles according to an embodiment of the present invention. The

solid arrows are associated with a-planes and the dashed arrows are associated with m-planes. For this diagram, the c-axis is normal to the plane of the figure. According to embodiments of the present invention, for the GaN wafer, the c-plane is (nearly) normal to wafer surface, so that the m-directions and a-directions are oriented as illustrated in FIG. 2B. It should be noted that the example illustrated in FIG. 2B utilizes one axis convention, but other axis conventions can be utilized, including an axis convention that is rotated 180° relative to the wafer flat. Both conventions are symmetrically equivalent.

The miscut target direction **250** is pointed towards the left (parallel to flat) for each of the axis conventions. The predetermined specification window **251** can be represented as a box that is centered on an m-direction, but also includes a-directions. For the illustrated axis convention, the miscut target is along  $[-1100]$ . For the alternative axis convention that is rotated 180° relative to the wafer flat, the miscut target is along  $[1-100]$ . Therefore, for both conventions, if the flat is on the bottom side, both miscut targets point to the left.

FIG. 2C is a schematic diagram illustrating the vector nature of the miscut angle according to an embodiment of the present invention. As illustrated in FIG. 2C, the miscut angle  $R$  is a vector quantity, defined by components in the m- and a-directions. Both can vary independently across a wafer, and may have different effects. Physically, this vector can be considered as the crystallographic c-direction projected onto the plane defined by the surface of the wafer, as shown by the square region **250**.

FIG. 2D is a schematic diagram illustrating an alternative specification for miscut angle according to an embodiment of the present invention. In FIG. 2D, the miscut specification **260**, rather than being specified by the miscut angle with respect to the m-plane and a-plane components, could be specified by the magnitude  $|R|$  and the direction  $\theta$  of the miscut angle, as shown by region **260**.

FIG. 3A is a Nomarski micrograph of epitaxial surface, for case when substrate misorientation is  $<0.15^\circ$ . FIG. 3B is a Nomarski micrograph of epitaxial surface, for case when substrate misorientation is  $>0.65^\circ$ . As illustrated in FIG. 3A, the surface morphology of epitaxial layers grown on a substrate having a misorientation less than  $0.15^\circ$  is characterized by large hexagonal hillocks, with lateral dimensions on the order of tens to hundreds of microns and heights of up to several microns. The inventors believe that the lateral dimensions of the hillocks increases as the thickness of the epitaxial layers increases. Referring to FIG. 3B, the surface morphology of epitaxial layers grown on a substrate having a misorientation greater than  $0.65^\circ$  is characterized by a scalloped surface, which can also be referred to as a ridged or fish-scale surface, with lateral and vertical dimensions on the order of up to several microns. For both of these substrate misorientation values outside the range included within the scope of the present invention, devices fabricating using epitaxial layers grown on these substrates, are characterized by undesirable levels of device leakage during operation in high power regimes.

As described in additional detail in relation to FIG. 4A, the inventors have determined that a strong correlation exists between the substrate misorientation, the surface morphology, and the high power device performance. For substrate misorientations of  $\sim 0.4^\circ$  to  $0.5^\circ$  from the m-direction, good surface morphology results, producing devices that have improved high power operating characteristics.

FIG. 4A is a plot showing the surface morphologies of various epitaxially grown layers mapped against substrate misorientation in orthogonal directions (i.e., the  $\langle 1\bar{1}00 \rangle$  and

$\langle 11\bar{2}0 \rangle$  directions). The vertical axis represents the miscut angle in degrees toward the a-direction (towards  $\langle 11\bar{2}0 \rangle$ ). The horizontal axis represents the miscut angle toward the m-direction (towards  $\langle 1\bar{1}00 \rangle$ ). As illustrated in FIG. 3, good morphology for the epitaxial layers (solid black circles) results when the miscut angle ( $\theta$ ) is in vicinity of zero miscut toward the a-direction and  $\sim 0.35^\circ$ – $0.55^\circ$  degrees miscut toward the m-direction. In a particular embodiment, a miscut angle toward the m-direction of between  $-0.4^\circ$ – $0.5^\circ$  is utilized.

It should be noted that holding the miscut angle of the substrate towards the a-direction at substantially zero degrees, good surface morphology can be obtained. Referring to FIG. 4A, it should also be noted that good surface morphology can also be obtained by adjusting both the misorientation with respect to the a-direction and the m-direction. As illustrated by the poor surface morphology for a-direction =  $-0.13^\circ$  and m-direction =  $-0.33^\circ$  (data point **421**) and the good surface morphology obtained for miscut towards the a-direction by  $-0.13^\circ$  and m-direction =  $-0.43^\circ$  (data point **423**), the increase in the absolute value of the miscut angle in the m-direction will also result in good surface morphology. Thus, variation in the miscut angle with respect to the a-direction can be compensated for by variation in the miscut angle with respect to the m-direction. One of ordinary skill in the art would recognize many variations, modifications, and alternatives.

Referring to FIG. 4A, 70 data points are illustrated, representing five points measured on 14 GaN substrates. Data provided in relation to the five points on each substrate included the miscut information. The miscut specification for these wafers was miscut toward the m-direction of  $-0.4^\circ$  and miscut toward the a-direction of  $0^\circ$ , both with a tolerance  $\pm 0.3^\circ$ . The five points were imaged using Nomarski microscopy and judged to be Good (solid circles), Bad (x) or Borderline (open circles). The Good morphologies were obtained for miscut towards the a-direction close to  $0^\circ$  and miscut toward the m-direction of  $\sim -0.4^\circ$ – $-0.5^\circ$ . Thus, as demonstrated by the data, a miscut with respect to the  $\langle 0001 \rangle$  direction of less than  $-0.3^\circ$  toward the m-direction results in poor morphology. Additionally, at high miscut angles (i.e., greater than  $\pm 0.6$  with respect to the  $\langle 0001 \rangle$  direction results in poor morphology.

Given the manufacturing tolerances and variation in crystal surface orientation across the wafer, embodiments of the present invention utilize a target miscut value that produces the largest possible area of material with good surface morphology (and consequent high device yield). As illustrated in FIG. 4A and discussed above, a larger miscut towards the m-direction can accommodate some variation in miscut towards the a-direction. Referring to FIG. 4A, an a-direction miscut of  $-0.13^\circ$  is represented by the horizontal dashed line. For this value of a-direction miscut, an m-direction miscut of less than  $0.35^\circ$  produces poor morphology, while a greater m-direction miscut results in good morphologies. In this manner, the m-direction and a-direction miscuts interact to affect the morphology, and a larger m-direction miscut can be utilized to accommodate variation in a-direction miscut.

It should be noted that although the substrate specifications may specify a particular miscut orientation for the substrate, the variation in orientation across the substrate may result in some regions of the substrate being characterized by a misorientation within the ranges provided by embodiments of the present invention and other regions characterized by a misorientation outside the ranges provided by embodiments of the present invention. In other words, substrate manufacturers allow for some margin in miscut variation. The variation in

miscut tends to be relatively large for pseudo-bulk GaN grown by HVPE on non-native substrates. Therefore the surface morphology will vary accordingly across the wafer, as indicated the data in FIG. 4A.

As an example, if the substrate specification is for a misorientation toward the m-direction of  $0.3^\circ \pm 0.3^\circ$ , then regions of the substrate can be characterized by zero misorientation while other regions are characterized by a misorientation of  $0.6^\circ$ . The inventors have determined that, for substrates with varying misorientation angles, the morphology is good in regions with a misorientation angle within the ranges of the embodiments described herein, which can be correlated to improved device performance.

FIG. 4B is a plot showing wafer quality data as a function of miscut angle according to an embodiment of the present invention. The wafer quality data was measured based on Nomarski images taken from wafers grown with various miscut angles. Each point represents a vector, beginning at the origin and ending at the datapoint. Wafers with "Good" quality are indicated by diamond symbols, wafers with "Bad" quality are indicated by square symbols, and wafers with borderline quality are indicated by triangle symbols. By considering the miscut toward the a-direction, the inventors have determined that poor morphology may be related to miscut angle as well as to the magnitude.

FIG. 5A is a plot showing the forward bias current-voltage characteristics of a high voltage PN diode according to an embodiment of the present invention. FIG. 5B is a plot showing the reverse bias current-voltage characteristics of a high voltage PN diode according to an embodiment of the present invention. As illustrated in FIG. 5A, the PN diode turns on at  $\sim 3\text{V}$  with a substantially linear I-V characteristic. Referring to FIG. 5B, under reverse bias, the PN diode conducts substantially no current until the voltage reaches  $\sim 2,500\text{V}$ , at which breakdown occurs. Thus, embodiments of the present invention are suitable for high voltage (e.g., greater than  $400\text{V}$ ) operation.

FIG. 6 is a plot showing the reverse bias current-voltage characteristics of a high-voltage GaN PN diode according to an embodiment of the present invention (solid curve) compared to a GaN PN diode fabricated using a conventional substrate, which can include substrates that are cut at misorientation angles ( $\theta$ ) outside of the ranges suitable for use according to embodiments of the present invention.

As illustrated in FIG. 6, under reverse bias, the reverse leakage current is substantially the same up to voltages of  $\sim 700\text{V}$ . At reverse bias voltages over  $700\text{V}$ , although the high voltage PN diode fabricated according to embodiments of the present invention maintains a substantially linear increase in reverse leakage current with voltage (plotted on a logarithmic scale). Thus, in devices, for example, PN diodes, Schottky diodes, vertical JFETs, HEMTs, integrated FETs and diodes, merged PN/Schottky diodes, and the like, that operate at high voltages, e.g.,  $>600\text{V}$ ,  $>1200\text{V}$ ,  $>1700\text{V}$ , or the like, the use of substrates with a predetermined misorientation angle provide improved performance, particularly in high voltage regimes.

It should be noted that embodiments of the present invention are particularly suitable for applications in high power regimes. For low power regimes, associated with some LED and laser operation, and other forward bias operation, or the like, the impact of the substrate not being miscut within the predetermined range provided by embodiments of the present invention would not be detectable, since, as illustrated in FIG. 6, effects dependent on the proper miscut angle are not observable in some implementations until high power regimes are entered. Thus, embodiments of the present inven-

tion are suitable for device applications utilizing thick epitaxial layers (e.g., a drift layer  $>3\text{ }\mu\text{m}$  thick) that are operated in high power regimes (e.g.,  $>200\text{V}$ ). Since conventional GaN devices operate in low power regimes (e.g., less than  $200\text{V}$ ), the impact of the proper misorientation of the substrate would not have been observed during typical operation. The inventors, on the contrary, have appreciated the scope of the problem presented by substrates oriented at conventional orientations during operating in high power regimes. Thus, embodiments of the present invention are particularly applicable to devices in which one or more of the epitaxial layers are characterized by low doping and high thickness, for example, the drift layer of a vertical PN GaN diode. Thus, embodiments of the present invention are particularly suitable for devices that include epitaxial layers over  $5\text{ }\mu\text{m}$  in thickness.

FIG. 7 is a simplified flowchart illustrating a method of fabricating an electronic device according to an embodiment of the present invention. Referring to FIG. 7, the method includes providing a III-V substrate having a hexagonal crystal structure and a normal to a growth surface characterized by a misorientation from the  $\langle 0001 \rangle$  direction of between  $0.15^\circ$  and  $0.65^\circ$  (710). The III-V substrate is an n-type GaN substrate in a specific embodiment. In an embodiment, the normal to the growth surface is misoriented towards toward the negative  $\langle 1\bar{1}00 \rangle$  direction, for example, in a range of between  $0.4^\circ$  and  $0.5^\circ$ . In a particular embodiment, the normal to the growth surface is characterized by a misorientation towards the  $\langle 11\bar{2}0 \rangle$  direction of substantially zero degrees. In other embodiments, the misorientation has components toward/away from both the  $\langle 1\bar{1}00 \rangle$  direction and the  $\langle 11\bar{2}0 \rangle$  direction.

The method also includes growing a first epitaxial layer coupled to the III-V substrate (712) and growing a second epitaxial layer coupled to the first epitaxial layer (714). For some devices, the method includes forming an isolation region disposed laterally to the second epitaxial layer. In some high power device applications, the first epitaxial layer comprises an n-type GaN epitaxial layer having a thickness greater than  $3\text{ }\mu\text{m}$  and the second epitaxial layer comprises a p-type GaN epitaxial layer. In some implementations, the method also includes forming a third epitaxial layer disposed between the second epitaxial layer and the second contact. A doping density of the third epitaxial layer is higher than a doping density of the second epitaxial layer.

Additionally, the method includes forming a first contact in electrical contact with the substrate (716) and forming a second contact in electrical contact with the second epitaxial layer (718). As an exemplary device, a PN diode can be fabricated using the techniques described herein, with the first contact being a cathode and the second contact being an anode of the PN diode. The electronic device can also be a Schottky diode.

It should be appreciated that the specific steps illustrated in FIG. 7 provide a particular method of fabricating an electronic device according to an embodiment of the present invention. Other sequences of steps may also be performed according to alternative embodiments. For example, alternative embodiments of the present invention may perform the steps outlined above in a different order. Moreover, the individual steps illustrated in FIG. 7 may include multiple sub-steps that may be performed in various sequences as appropriate to the individual step. Furthermore, additional steps may be added or removed depending on the particular applications. One of ordinary skill in the art would recognize many variations, modifications, and alternatives.

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It is also understood that the examples and embodiments described herein are for illustrative purposes only and that various modifications or changes in light thereof will be suggested to persons skilled in the art and are to be included within the spirit and purview of this application and scope of the appended claims.

What is claimed is:

1. An electronic device comprising:  
a III-V substrate having a hexagonal crystal structure and a normal to a growth surface characterized by a misorientation from the  $\langle 0001 \rangle$  direction of between  $0.15^\circ$  and  $0.65^\circ$ ;  
a first III-V epitaxial layer coupled to the III-V substrate;  
a second III-V epitaxial layer coupled to the first III-V epitaxial layer;  
a first contact in electrical contact with the III-V substrate; and  
a second contact in electrical contact with the second III-V epitaxial layer.
2. The electronic device of claim 1 wherein the normal to the growth surface is misoriented towards the negative  $\langle 1\bar{1}00 \rangle$  direction.
3. The electronic device of claim 1 wherein the misorientation is between  $0.4^\circ$  and  $0.5^\circ$ .
4. The electronic device of claim 1 wherein the normal to the growth surface is characterized by a misorientation towards the  $\langle 11\bar{2}0 \rangle$  direction of substantially zero degrees.

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5. The electronic device of claim 1 wherein the normal to the growth surface is misoriented towards the  $\langle 1\bar{1}00 \rangle$  direction by an angle ranging between  $-0.4^\circ$  and  $-0.5^\circ$  and towards the  $\langle 11\bar{2}0 \rangle$  direction by an angle ranging between  $-0.1^\circ$  and  $-0.2^\circ$ .

6. The electronic device of claim 1 wherein the III-V substrate comprises an n-type GaN substrate.

7. The electronic device of claim 1 wherein the first III-V epitaxial layer comprises an n-type GaN epitaxial layer and the second III-V epitaxial layer comprises a p-type GaN epitaxial layer.

8. The electronic device of claim 1 wherein the electronic device comprises a PN diode, the first contact comprises a cathode of the PN diode, and the second contact comprises an anode of the PN diode.

9. The electronic device of claim 1 further comprising an isolation region disposed laterally to the second III-V epitaxial layer.

10. The electronic device of claim 1 further comprising a third III-V epitaxial layer disposed between the second III-V epitaxial layer and the second contact, wherein a doping density of the third III-V epitaxial layer is higher than a doping density of the second III-V epitaxial layer.

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